

# Synthetic paramontroseite VO<sub>2</sub> with good aqueous lithium-ion battery performance†

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**Synthetic paramontroseite VO<sub>2</sub> has been successfully obtained using a simple chemical reaction route for the first time after fifty years; the paramontroseite phase shows a conducting property and good aqueous lithium ion battery performance.**

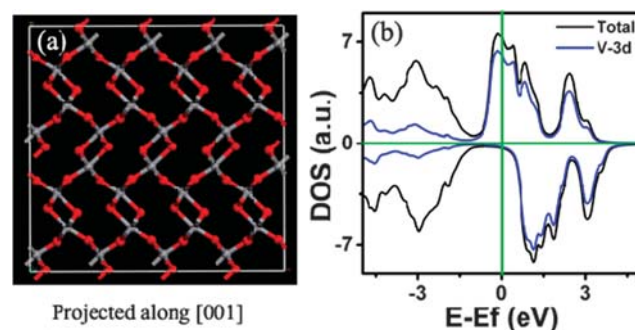
In the 1950s, Mrose and Evans discovered a new mineral paramontroseite VO<sub>2</sub> from the Colorado Plateaus region.<sup>1</sup> Paramontroseite vanadium dioxide (VO<sub>2</sub>) possesses a regular tunnel form, in which the VO<sub>6</sub> octahedra are linked into double chains and the double chains link corners with each other in turn to form a framework having tunnels with rectangular-shaped cross sections that are 1×2 octahedra on a side.<sup>2,3</sup> For more than fifty years, efforts have been strongly focused on the structural analysis of paramontroseite and the transformation relationship between paramontroseite and montroseite, owing to its fascinating tunnelled structure.<sup>1–3</sup> However, almost all of the experimental data for paramontroseite originated from investigation of mineral samples, and not synthetic samples. Here, we highlight an available pathway to accomplish this challenge by a simple reaction of sodium orthovanadate (Na<sub>3</sub>VO<sub>4</sub>·12H<sub>2</sub>O) and thioacetamide (TAA).

The fascination for paramontroseite VO<sub>2</sub> also arises from its structural analysis and calculation results, which indicate potentially conducting behavior. For the paramontroseite VO<sub>2</sub> phase, due to the presence of infinite chains along the *c*-axis with nearest V–V distances of 0.293 nm, its d-orbital electrons would be shared by all of the metal V atoms along the paramontroseite *c*-axis similar to that in tetragonal rutile VO<sub>2</sub><sup>4</sup> (see ESI,† S1). Therefore, paramontroseite would appear to have promising electrical conducting properties at room temperature from the structural point of view. Here, first-principle calculations using the VASP program were also performed to further understand the conducting behavior of paramontroseite VO<sub>2</sub>, and the spin-dependent density of states (DOS) and energy bands of paramontroseite VO<sub>2</sub> are shown in Fig. 1. In Fig. 1(b), the DOS resides across the Fermi level and gives a certain local DOS value at the Fermi level, indicating its metallic character. In more detail, the total DOS value of paramontroseite VO<sub>2</sub> around the Fermi level is mainly contributed from the V 3d and O 2p orbitals as shown in Fig. 1(b).

Further, the paramontroseite phase shows a continuous DOS starting from –0.93 eV and the Fermi level tangles with its conducting bands for all the high-symmetry points (see ESI,† S2). Comparing these calculation results with those for rutile and monoclinic VO<sub>2</sub>, it is found that the DOS of rutile VO<sub>2</sub> also possesses a continuous DOS around the Fermi level while the DOS of monoclinic VO<sub>2</sub> has a 0.1 eV wide deep minimum just above  $\epsilon_F$  with a separated band.<sup>5</sup> Therefore, the characteristics of DOS and energy bands for paramontroseite VO<sub>2</sub> are quite similar to those of the metallic rutile VO<sub>2</sub> phase, rather than that of the semiconducting monoclinic VO<sub>2</sub> phase. This similarity of calculation results between the paramontroseite VO<sub>2</sub> and metallic rutile VO<sub>2</sub> phase thus provides further evidence for the metallic behavior of the paramontroseite VO<sub>2</sub> structure.

Moreover, as observed from the DOS and energy bands for paramontroseite VO<sub>2</sub> (see ESI,† S2), it can be seen that the one spin direction (spin up) behaves like a metal and the other (spin down) is semiconducting, showing the half-metal property with a complete polarization of electrons at the Fermi level. That is to say, paramontroseite VO<sub>2</sub> in orthorhombic structure is also amenable to spintronics and spin injection.<sup>6</sup>

Our experimental results show that the paramontroseite VO<sub>2</sub> could be chemically synthesized by a simple reaction of sodium orthovanadate (Na<sub>3</sub>VO<sub>4</sub>·12H<sub>2</sub>O) and thioacetamide (TAA). Details of the experimental process can be described as follows: first, 4 mmol Na<sub>3</sub>VO<sub>4</sub>·12H<sub>2</sub>O and 20 mmol thioacetamide (TAA) were dissolved in 50 ml distilled water to form a homogeneous solution in a glass jar. This solution was then loaded into a 50 ml Teflon-lined autoclave, which was sealed and heated at 220 °C for 10 h. The system was then allowed to



**Fig. 1** (a) The supercell model of orthorhombic paramontroseite VO<sub>2</sub> projected along [001]. (b) Spin-dependent density of states. The black line shows the total DOS, the blue lines the V 3d partial DOS.

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† Electronic supplementary information (ESI) available: Structural analysis, band gap, experimental details, morphology, SAED pattern and XPS analysis of the sample. See DOI: 10.1039/b806009f

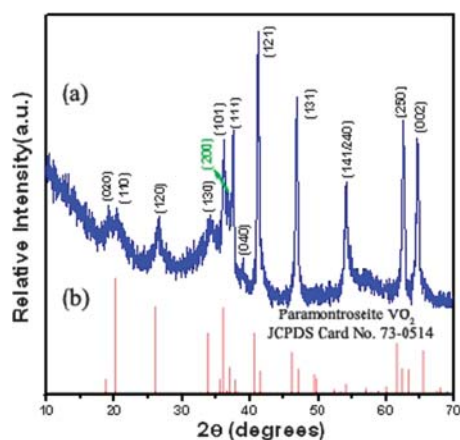


Fig. 2 XRD pattern for the paramontroseite VO<sub>2</sub>.

cool to room temperature. The final product was collected by centrifugation, and washed with deionized water and ethanol to remove any possible ionic remnants, and then dried in vacuum at 60 °C.

The XRD pattern of the as-obtained product could readily be indexed to orthorhombic paramontroseite VO<sub>2</sub> with the cell parameters  $a = 4.956$  Å,  $b = 9.332$  Å and  $c = 2.896$  Å, as shown in Fig. 2.† Further evidence for the presence of paramontroseite VO<sub>2</sub> comes from a high-resolution TEM image (HRTEM) and selected area electron diffraction (SAED) pattern (Fig. 3). Since paramontroseite VO<sub>2</sub> exhibits a wal-

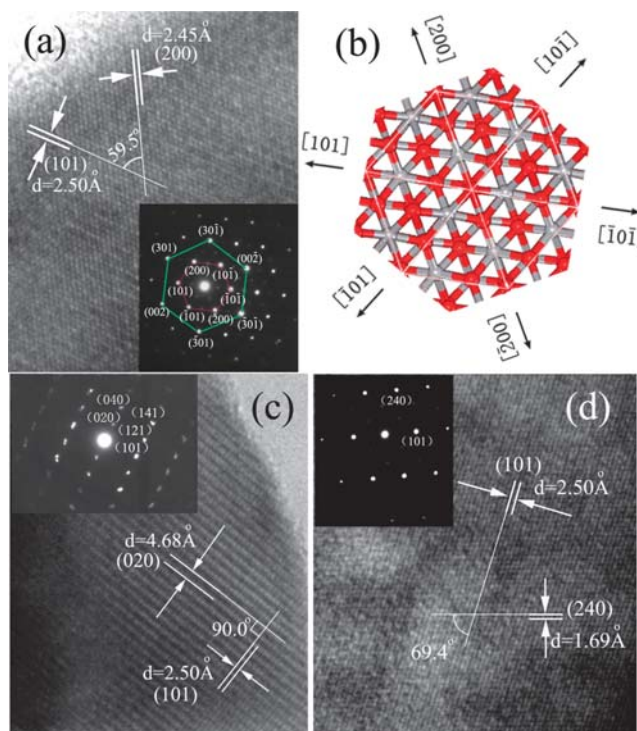


Fig. 3 High-resolution TEM image and selected electron diffraction patterns on the edge of a typical paramontroseite VO<sub>2</sub> walnut-like architecture. (b) Approximate hexagonal atomic arrangement in the paramontroseite crystal structure when projected along the  $b$ -axis, further conforming the as-obtained product was paramontroseite VO<sub>2</sub>.

nut-like morphology according to the panoramic FE-SEM image (see ESI,† S4), the HRTEM images and its corresponding SAEDs were performed on the edges of the typical VO<sub>2</sub> walnut-like architectures as shown in Fig. 3. The interplanar distances of 2.50, 2.45, 4.68, 1.69 Å match well with the  $d_{101}$ ,  $d_{200}$ ,  $d_{020}$ ,  $d_{240}$  spacings, respectively, of paramontroseite VO<sub>2</sub>. Moreover, based on the careful calculations for the angle values of each lattice plane in the same SAED pattern, it is found that all the orientation angle values of these planes appeared in the SAED patterns were fairly consistent with those calculated from orthorhombic crystallographic parameters of paramontroseite VO<sub>2</sub> (see ESI,† S5), which provided further clear evidence for the formation of crystallographic paramontroseite in the as-obtained sample.

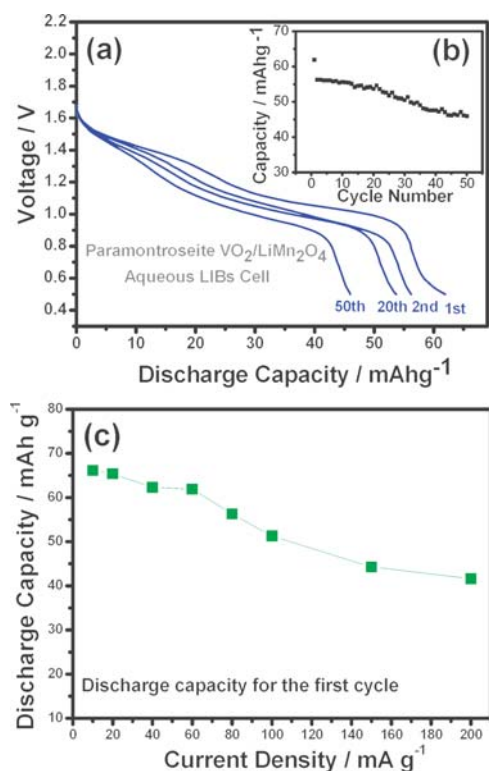
Of note, the crystal structure of paramontroseite VO<sub>2</sub> as shown in Fig. 3(b) reveals a unique approximate-hexagonal atomic arrangement when projected along the  $b$ -axis direction. Such approximate-hexagonal symmetry had already been theoretically recognized in Mader's calculated ED pattern.<sup>7</sup> Our experimental ED pattern shown in Fig. 3(a) also gives the approximate-hexagonal characteristic and the crystallographic plane indexes agrees well with those in the calculated ED pattern, which provided further diagnostic and direct evidence for the paramontroseite phase.

The room-temperature conductivity of the as-obtained paramontroseite VO<sub>2</sub> was measured to be 1.02 S cm<sup>-1</sup>, showing higher conductivity than many other inorganic conducting materials.<sup>8–10</sup>

The as-obtained conducting paramontroseite VO<sub>2</sub> walnut-like nanoarchitectures should thus facilitate the intercalation/deintercalation of lithium ions in lithium ion batteries (LIBs) due to the high electrical conductivity and the increased diffusion coefficient within the nanoarchitectures.<sup>11,12</sup> The tunnel size (4.946 Å × 2.851 Å) in the supercell crystal structure of paramontroseite VO<sub>2</sub> is considerably larger than the diameter of Li ion (1.36 Å). Obviously, the presence of the tunnels in paramontroseite VO<sub>2</sub> could be effective in facilitating Li-ion diffusion through the crystal structure, so facilitating the electrochemical reaction, which should contribute the Li-ion insertion performance in the Li-ion battery. Bearing in mind that the above advantage of paramontroseite VO<sub>2</sub>, here, we introduced paramontroseite VO<sub>2</sub> as the active material for aqueous lithium ion battery applications, on the consideration that the development of an aqueous solution as electrolyte in rechargeable lithium-ion batteries was an effective way to realize a highly desirable<sup>13</sup> safer and less-expensive battery by replacement of organic electrolyte.

The discharge curves of the paramontroseite VO<sub>2</sub>/LiMn<sub>2</sub>O<sub>4</sub> aqueous LIB (5 M LiNO<sub>3</sub> and 0.001 M LiOH) are shown in Fig. 4(a) which displays the 1st, 2nd, 20th and 50th cycle of the cell with cutoff voltages of 1.70 and 0.5 V at a current density of 60 mA g<sup>-1</sup>.

It can be clearly found that the paramontroseite VO<sub>2</sub> electrode promotes the electrochemical performance in aqueous LIBs. First, the paramontroseite VO<sub>2</sub> electrode shows both discharge capacity and the output voltage advantages in our aqueous LIBs, revealing the high energy density of the paramontroseite VO<sub>2</sub>/LiMn<sub>2</sub>O<sub>4</sub> cell. The first discharge capacity of the VO<sub>2</sub> electrode in our paramontroseite



**Fig. 4** (a) Voltage vs. discharge capacity curves for paramontroseite VO<sub>2</sub> electrode in aqueous 5 M LiNO<sub>3</sub> and 0.001 M LiOH as electrolyte for the 1st, 2nd, 20th and 50th cycle. The inset (b) in (a) is the cycling behavior of the paramontroseite VO<sub>2</sub>/LiMn<sub>2</sub>O<sub>4</sub> aqueous cell. (c) The first discharge capacity at different current densities.

VO<sub>2</sub>/LiMn<sub>2</sub>O<sub>4</sub> aqueous cell is 61.9 mAh g<sup>-1</sup>, which is a highly significant value for aqueous LIB systems in the light of previous reports,<sup>14–16</sup> although it is still smaller than that in the organic electrolytes (usually > 120 mA h g<sup>-1</sup>).<sup>17</sup> Furthermore, the first output voltage of this aqueous LIB is in the range of 1.3–1.6 V, while the second average output voltage is 1.0–1.2 V. Here, the output voltage is again very impressive when compared with previous aqueous LIBs.<sup>14–16</sup> Also, the first output voltage for paramontroseite VO<sub>2</sub>/LiMn<sub>2</sub>O<sub>4</sub> aqueous LIB is higher than the output voltage of primary Zn–MnO<sub>2</sub>, Ni–MH and Ni–Cd cell systems (*ca.* 1.2 V), while even the lowest output voltage approached 1.2 V. The relatively high output voltage and high discharge capacity value infer that our aqueous LIB is the promising substitute candidate for cells with higher energy density.

Second, it is clear that the as-obtained paramontroseite VO<sub>2</sub> electrode exhibits better cycling behavior than those in the previously reported aqueous LIBs.<sup>14–16</sup> Fig. 4(b) (inset in (a)) shows the cycling behavior at a current density of 60 mA g<sup>-1</sup> between 0.5 and 1.7 V in aqueous electrolyte. It is found that the discharge capacity after 50 cycles is 45.96 mAh g<sup>-1</sup>, which is approximately 74% of the first discharge capacity, while it shows good resistance to fading up to 50 cycles. Third, the paramontroseite VO<sub>2</sub> electrode exhibits a good stability at high current density as shown in Fig. 4(c), showing the

relatively high-rate performance of this electrode, which resulted from the mitigation of kinetic limitations by the high electronic conductivity of paramontroseite VO<sub>2</sub>.

In conclusion, synthetic paramontroseite VO<sub>2</sub> has been successfully obtained in this work by a simple chemical reaction route after more than 50 years; previously the chemical and physical information of paramontroseite only came from mineral samples. Both structural analysis and the first-principle calculations reveal the conducting property of paramontroseite VO<sub>2</sub> and experiments confirm its good conductivity. The as-obtained paramontroseite sample shows the fascinating characteristics of tunnelled crystal structure, nanoarchitected morphology and its conducting property, which entail superior aqueous LIB performance in the VO<sub>2</sub>/LiMn<sub>2</sub>O<sub>4</sub> aqueous LIB system. Compared with previous aqueous LIBs, the paramontroseite VO<sub>2</sub>/LiMn<sub>2</sub>O<sub>4</sub> cell exhibits high energy density, better cycling behavior and high-rate capability, which is of great use for potential industrial applications.

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## Notes and references

† The sample was characterized by X-ray powder diffraction (XRD) with a Philips X'Pert Pro Super diffractometer with Cu-K<sub>α</sub> radiation ( $\lambda = 1.54178 \text{ \AA}$ ). The field emission scanning electron microscopy (FESEM) images were taken on a JEOL JSM-6700F SEM. The transmission electron microscopy (TEM) images were carried out on a JEOL-2010 TEM at an acceleration voltage of 200 kV. The electrical conductivity of the as-prepared paramontroseite VO<sub>2</sub> was measured with the use of a computer-controlled, four-probe technique.

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